

## Understanding tetrahedral liquids through patchy colloids

Ivan Saika-Voivod, Frank Smallenburg, and Francesco Sciortino

Citation: *The Journal of Chemical Physics* **139**, 234901 (2013); doi: 10.1063/1.4840695

View online: <http://dx.doi.org/10.1063/1.4840695>

View Table of Contents: <http://scitation.aip.org/content/aip/journal/jcp/139/23?ver=pdfcov>

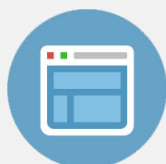
Published by the [AIP Publishing](#)

---



## Re-register for Table of Content Alerts

Create a profile.



Sign up today!



## Understanding tetrahedral liquids through patchy colloids

Ivan Saika-Voivod,<sup>1</sup> Frank Smallenburg,<sup>2</sup> and Francesco Sciortino<sup>2</sup>

<sup>1</sup>*Department of Physics and Physical Oceanography, Memorial University of Newfoundland, St. John's, Newfoundland A1B 3X7, Canada*

<sup>2</sup>*Dipartimento di Fisica, Sapienza Università di Roma, Piazzale Aldo Moro 5, 00185 Roma, Italy*

(Received 9 September 2013; accepted 22 November 2013; published online 16 December 2013)

We investigate the structural properties of a simple model for tetrahedral patchy colloids in which the patch width and the patch range can be tuned independently. For wide bond angles, a fully bonded network can be generated by standard Monte Carlo or molecular dynamics simulations of the model, providing a good method for generating defect-free random tetrahedral networks. This offers the possibility of focusing on the role of the patch angular width on the structure of the fully bonded network. The analysis of the fully bonded configurations as a function of the bonding angle shows how the bonding angle controls the system compressibility, the strength of the pre-peak in the structure factor, and ring size distribution. Comparison with models of liquid water and silica allows us to find the best mapping between these continuous potentials and the colloidal one. Building on previous studies focused on the connection between angular range and crystallization, the mapping makes it possible to shed new light on the glass-forming ability of network-forming tetrahedral liquids.

© 2013 AIP Publishing LLC. [<http://dx.doi.org/10.1063/1.4840695>]

### I. INTRODUCTION

Cooling a liquid below its melting temperature will result in either crystallization or vitrification.<sup>1</sup> Understanding on a basic level why one alternative is predominantly chosen over the other in different systems remains fundamentally an open problem. This is particularly acute in tetrahedral network-forming liquids such as water and silica. In both liquids, corner-sharing tetrahedral units form a random network that fills space. However, in nature, silica is the prototypical glass former, while water crystallizes readily. Moreover, different models of water used in simulation vary in their ability to crystallize.

Recent advances in the synthesis of colloidal particles have brought about a physical realization of the patchy particle.<sup>2,3</sup> Of particular relevance as models of networked liquids are colloids with sticky spots that provide strongly directional interactions with fixed valence.<sup>4–6</sup> To better understand the physics of self-assembly in these systems, theorists have been studying colloidal particles with simple models that incorporate information concerning the valence. These models can be considered as an evolution of primitive models for associated liquids.<sup>7–9</sup> Particularly relevant for our investigation is the Kern-Frenkel (KF) model,<sup>10</sup> a model which has its origin in the work of Bol<sup>11</sup> and that offers a very clean separation between the angular width and radial extent of the attractive interaction between patches.

Recent systematic simulation studies of the KF model with tetrahedrally arranged patches have uncovered the dependence of both the driving force for nucleation, namely, the chemical potential difference between crystal and liquid, and the free energy barriers to nucleation on the width of the patches.<sup>12–14</sup> More recently, the KF model was extended to retain its tetravalent character even in the case where patches are wide enough to accommodate more than one bond. Extending

the model to guarantee the single-bond-per-patch condition regardless of patch width, under appropriate conditions, results in the liquid retaining thermodynamic stability down to zero temperature, avoiding the crystal phase altogether.<sup>15</sup>

The relative simplicity of the KF model allows us to understand its thermodynamic properties through bonding entropy in geometric terms, at least for the crystal. For the liquid, the computational efficiency with which the model can be simulated allows for detailed calculations of contributions to the free energy down to states at or near the ground state, which has a well defined energy. Beyond being a model for patchy colloids, the KF model provides a potential coarse grained description of bond flexibility in more general network-forming systems. One requires a way of mapping systems onto the KF model with particular patch widths.

In particular, if one is able to provide such a map for different network-forming liquids, then one can unlock the insights we have gained from the patch width dependence of crystallization in KF to inform our understanding of why, for example, water freezes while silica does not.

In the present work, we compare the networks formed in BKS silica,<sup>16</sup> ST2 water,<sup>17</sup> TIP4P/2005 water,<sup>18</sup> mW water,<sup>19</sup> and Stillinger-Weber (SW) silicon.<sup>20</sup> We do so always at the optimal network-forming density for each system,<sup>21</sup> thus eliminating density as a parameter, and find markers of network formation that allow us to make a mapping of these systems onto KF models with different patch widths. We focus on the best available network configurations across the systems. For the KF model, we utilize only fully bonded networks.

The models represent different classes in the way that tetrahedral geometry is enforced. For KF, mW, and SW, the geometry is enforced at the level of bonding between particles, whether through the placement of patches, or through a

three-body interaction term. ST2 and TIP4P/2005 have tetrahedrality built into the rigid shape of the water molecule through charge sites, which in turn direct hydrogen bonding. The BKS model is a binary mixture, with bridging between Si ions mediated by relatively large and soft oxygen ions that arrange themselves tetrahedrally, essentially through stoichiometry and steric repulsion. All of these systems can be understood in terms of the random network model,<sup>22–28</sup> although the perturbations required to quantitatively account for the particularities of each system are potentially qualitatively different. Given these differences, it is not obvious that a mapping to the KF model via the patch width, a single parameter, is a realistic endeavor from the outset.

Mapping of various network-forming systems to other models where bond flexibility can be tuned, such as the mW–SW family of models, could prove a worthwhile pursuit. The work of Molinero *et al.*<sup>29</sup> on the dependence of the location of the liquid-liquid critical point in the model on the strength of the three-body interaction term lays the groundwork for such a study. However, the extent to which crystallization is understood in the KF model makes it an ideal candidate for unlocking insights into other network-forming liquids.

Additionally, we explore the ability of fully bonded KF networks to generate amorphous ground states for silica and water. A long-standing issue within the scientific community is the development of algorithms that can generate ideal random tetrahedral networks.<sup>30–32</sup> Thus, it is of interest to see how perfection in the KF model transfers, from an energetic perspective, to models for such emblematic network formers such as silica and water.

## II. METHODS

In this work, we compare the properties of the best available tetrahedral networks from several model liquids: the KF model for tetravalent patchy colloids, the BKS model of silica, the ST2, TIP4P/2005 and mW models of water, and the SW model of silicon. To facilitate comparison, structural quantities for all models are reported at or near the optimal density<sup>21</sup> or pressure for the formation of a bonded network. Generally, we identify this optimal condition with a local minimum in low temperature isotherms of potential energy. Further, all structural quantities are reported for configurations that are first quenched through a conjugate-gradient (CG) algorithm to a local minimum in the potential energy, i.e., to a so-called inherent structure (IS).<sup>33–35</sup> This eliminates disorder due to vibrations.

### A. KF model

The KF model we consider consists of hard sphere particles of diameter  $\sigma$ , each decorated with four tetrahedrally arranged patches. Each patch is defined by a cone with apex at the particle centre and aperture  $2\theta$ , where  $\theta$  is the angle between the cone axis and a generatrix. Particles form bonds of energy  $-\epsilon$  when patches overlap. Overlap occurs when the center-to-center distance between two particles is less than  $(1 + \delta)\sigma$ , with  $\delta = 0.12$ , and the line segment connecting parti-

cle centres passes through both patches, i.e., when the angle between the line segment and each of the patch axes is less than  $\theta$ . See Ref. 12 for details.

For sufficiently wide patches, i.e., for  $2\sin\theta > (1 + \delta)^{-1}$  (in our case  $\theta > 26.5^\circ$ , or  $\cos\theta < 0.895$ ), it becomes possible for a patch on one particle to form bonds simultaneously with more than one particle. To avoid this scenario, we enforce single bonding per patch through the extension of the KF model presented in Ref. 15 (and associated supplementary information), where two of us showed that for sufficiently wide patches over an appropriate range in density, the liquid state is more stable than the crystal down to  $T = 0$ . Briefly, the single-bond-per-patch condition is enforced by allowing bonds to form between overlapping patches subject to a probability based on the Boltzmann distribution. When more than one overlap occurs with a patch, only one of the overlaps, selected at random, is allowed to form a bond. Thus, for all patch widths the ground state energy per particle is  $-2\epsilon$ . For  $\cos\theta \geq 0.895$ , where the narrowness of the patches naturally restricts patches to only share a single bond, the extended KF model maps onto the original KF model with a rescaled  $\epsilon$ , or equivalently, at a rescaled  $T$ . At low  $T$ , the two models become equivalent.

The single-bond-per-patch condition is implemented in the event-driven molecular dynamics (EDMD) simulations of the model by introducing an event that randomly selects a bond to be re-evaluated according to the Boltzmann criterion employed to define bonds. The events are scheduled at random according to an exponential distribution with an average rate that is sufficiently high to ensure that the system dynamics are invariant with respect to the re-evaluation frequency. As all bonds have the same energy, and are therefore equally likely, it is always possible to reverse a bond reassignment in the next step with the same probability, ensuring detailed balance. See Ref. 15 again for details.

We also carry out Monte Carlo (MC) simulations of the original KF model (without enforced single bonding), as in Ref. 12, for  $\cos\theta \geq 0.895$ , as well as for the modified KF model with single bonding enforced. We find that MC is more efficient at equilibrating the systems at low  $T$  as the patches become narrow. Additionally, these simulations serve as a consistency check on the EDMD simulations.

Regardless of the simulation method, we report structural quantities only for fully bonded KF configurations, i.e., perfect networks at the ground state energy, at a density of  $\rho\sigma^3 = 0.57$ . Obtaining perfect networks becomes progressively more computationally challenging as  $\cos\theta$  increases, and we are not able to obtain perfect liquid configurations beyond  $\cos\theta = 0.92$ . For  $\cos\theta = 0.85, 0.87, 0.895$ , and  $0.92$ , we obtain only a single perfect configuration; these perfect narrow-patch configurations are obtained through MC.

For these narrower patch angles, multiple MC simulations for each angle ran for a total CPU time of six months to a year before a perfect configuration was found. The strong slow-down for narrow patches can be understood when considering the mechanism behind network reorganization in these systems. For narrow patches, bond rearrangement without energy penalty (via the bond re-evaluation moves) becomes less and less likely as the patches become smaller,

and impossible around  $\cos \theta = 0.89$ . Thus, bond breaking is required to reorganize the network. However, the small patch size also indicates a large entropy cost to forming a bond, and as a result the temperature at which (nearly) all bonds are formed is very low, ensuring that bond breaking events are rare. Additionally, unless another free bond is nearby, most broken bonds will reform almost instantaneously. Reaching a fully bonded state requires the sampling of a large number of network reorganizations, leading to extremely long simulation times.

## B. BKS silica

For BKS silica, we simulate 444  $\text{SiO}_2$  units in a cubic box of length 2.65761 nm, i.e., at a constant density of 2.36  $\text{g/cm}^3$ . At this density at low  $T$ , the tetrahedral network is well formed,<sup>36</sup> and this density is in fact near the minimum in potential energy of BKS silica as a function of density at  $T = 3000$  K.<sup>37,38</sup> Constant  $T$  molecular dynamics simulations from  $T = 3000$  to 2400 K are carried out with a time step of 1 fs using Gromacs software version 4.5.5<sup>39–42</sup> employing the Nosé-Hoover thermostat with a time constant of 1 ps and a radial cutoff of 1 nm for all real space pair interactions. Coulomb interactions are handled with the particle mesh Ewald (PME) algorithm with a fourier spacing of 0.1 nm and interpolation of order four (cubic). We add to the BKS potential a short range interaction, described in Ref. 37, to prevent the system from exploring the unphysical attraction occurring at small distances. At 2400 K, we simulate for 2.4  $\mu\text{s}$  as the system dynamics are quite slow, with Si ions diffusing a root mean square distance of approximately 0.35 nm (just over one average Si–Si distance) in 1  $\mu\text{s}$ . We use configurations from  $T = 2400$  K for structural analysis.

In order to facilitate comparison of the IS energy  $e_{\text{IS}}$  with previous work, in performing the quench we employ the slightly modified form of BKS employed in Ref. 37, in which the real space part of the potential tapers smoothly starting at 0.77476 nm from the original BKS value to zero at 1 nm. This procedure recovers the previously reported IS energies in Ref. 37 for the  $T$  overlapping with this work (2700 K to 3000 K), as shown in Fig. 6(d).

## C. Water and silicon models

For our simulations of the TIP4P/2005 model of water, we also use Gromacs v4.5.5 to carry out a series of  $NVT$  simulations, varying the number of particles from 231 to 309 and keeping the cubic box fixed at  $V = (2 \text{ nm})^3$ , thus varying density from approximately 0.86 to 1.16  $\text{g/cm}^3$ . Following Ref. 43, we employ a time step of 2 fs, a Nosé-Hoover thermostat with a time constant of 1 ps and a real space potential cutoff of 0.85 nm. For Coulomb terms, we use the PME algorithm with fourier spacing of 0.1 nm and interpolation order four (cubic). Initial configurations are equilibrated at 235 K before running at  $T = 193$  K. At the dynamically slowest state points (lowest density), the relaxation time, i.e., the time required to diffuse an average O–O distance, is very long

and on the order of 10  $\mu\text{s}$ . Thus, while our simulation times of a little over 20  $\mu\text{s}$  are not sufficient to obtain very precise averages at low density, they are sufficient to reach a diffusive steady state and to discern a minimum in the energy as a function of density near 0.92  $\text{g/cm}^3$ . For structural analysis, we therefore harvest configurations from our simulation containing 245 molecules, which corresponds to this optimal density.

Configurations for ST2 water (employing the reaction field treatment of electrostatic interactions) are taken from a density of 0.83  $\text{g/cm}^3$  (222 water molecules in a box of length 2 nm) at  $T = 235$  K. To obtain these low  $T$  configurations, we extend the successive umbrella sampling grand canonical Monte Carlo simulations of ST2 described in Ref. 44.

To extend our analysis to another class of tetrahedral liquid models, we perform a less thorough investigation of two members of the SW silica model family, that have different values of the parameter controlling the strength of the three-body, tetrahedrality-enforcing interaction term. We thank colleagues Vishwas Vashist and Srikanth Sastry for providing us with quenched configurations of the SW model with 10 000 particles at  $T = 1196$  K and  $P = -1.88$  GPa. The state point is above the liquid-liquid critical temperature for that model, but below the line of compressibility maxima and in the tetrahedral network regime.<sup>45</sup> We are also grateful to colleagues Dr. Flavio Romano and Dr. John Russo for providing us, in the course of their current study of the model, with quenched configurations of 686 mW water molecules at  $P = 0$  and  $T = 171.3$  K, a state point below the transition temperature to the low density liquid (LDL) as presented in Ref. 19, but at which crystallization also occurs.

## D. Structural quantities

We report the structure factor  $S(q)$ , defined as  $\langle \rho_{\vec{q}} \rho_{-\vec{q}} \rangle / N$ , where  $\rho_{\vec{q}} = \sum_{i=1}^N \exp(-i\vec{q} \cdot \vec{r}_i)$ ,  $\langle \dots \rangle$  denotes an ensemble average over reciprocal space vectors  $\vec{q}$  having magnitude  $q$  and  $N$  is the number of *node* particles in the systems. A node particle is simply the particle that is at the centre of a tetrahedron in the corner sharing tetrahedral network: for the KF, mW, and SW models, each particle is a node particle; for silica, the Si ions are node particles, while O atoms are the node particles for ST2 and TIP4P/2005 water. We also calculate the radial distribution function  $g(r)$  for node particles.

The other key structural quantity we present is the distribution of the angle  $\phi$ , defined by the vectors emanating from a node particle to two of its neighboring node particles (the Si–Si–Si angle for BKS and SW, O–O–O angle for water). For the continuous potential models, we define neighbors of a particle as those that lie within a radius of  $r_{\text{cut}}$ , determined from the minimum between first and second neighbor peaks in  $g(r)$ . Values of  $r_{\text{cut}}$  are given in Table I. For the KF model, neighbors are defined as particles sharing a bond.

Additionally, we report on the distribution of the sizes of minimal closed rings of neighboring node particles, with neighbors defined as above for the bond angle distribution. The algorithm for determining ring statistics is from Ref. 46.

TABLE I. Summary of parameters characterizing the systems studied:  $r_{\text{peak}}$  is the position of the first peak in  $g(r)$ ;  $\rho_{\text{eff}}^*$  is the value of the reduced density  $\rho r_{\text{peak}}^3$  at which structural analysis is reported;  $r_{\text{cut}}$  is the position of the first minimum in  $g(r)$ , and is used to determine neighbors;  $\bar{n}_b$  is the average number of neighbors;  $f_{\text{def}}$  is the fraction of particles that do not have four neighbors. For mW  $\sigma = 0.23925$  nm and for SW  $\sigma = 0.20951$  nm.

Model	$r_{\text{peak}}$	$\rho_{\text{eff}}^*$	$r_{\text{cut}}$	$\bar{n}_b$	$f_{\text{def}}$
KF	$1.06\sigma$	0.68	n.a.	4	0
BKS	0.315 nm	0.74	0.329 nm	4.002	0.0024
ST2	0.283 nm	0.63	0.320 nm	3.988	0.018
TIP4P/2005	0.2775 nm	0.65	0.315 nm	3.996	0.0046
mW	$1.143\sigma$	0.66	$1.40\sigma$	3.63	0.327
SW	$1.165\sigma$	0.70	$1.41\sigma$	3.50	0.475

### III. RESULTS

#### A. $S(q)$ and $g(r)$

In Figs. 1(a) and 1(b), we present  $S(q)$  and  $g(r)$  for the tetrahedral KF model with patch angles ranging from  $\cos \theta = 0.60$  ( $\theta = 53.1^\circ$ ) to  $\cos \theta = 0.92$  ( $\theta = 23.1^\circ$ ). The perfectly bonded configurations from which the curves are derived are obtained from low  $T$  simulations which sample the energetic ground state. Beyond  $\cos \theta = 0.82$ , the computational effort required to sample such states grows considerably. As  $\cos \theta$  increases, i.e., as the patch angle decreases, the peak characteristic of a structured network develops.<sup>47</sup> First a shoulder appears, and by  $\cos \theta = 0.82$  a peak in the form of a local maximum is established. Beyond  $\cos \theta = 0.82$ , the position  $q_1$  of this network peak does not change significantly, while its height  $S(q_1)$  grows (as does the height of the main peak at  $q_2 \approx 8$ ). Accompanying the emergence and growth of the network peak in  $S(q)$  is a significant decrease of the system compressibility, as evidenced by the approach toward zero of  $S(0)$ . Small values of  $S(0)$  are commonly understood to signify high degrees of hyperuniformity.<sup>48</sup> It is interesting to observe in passing that for the case of  $\cos(\theta) = 0.92$ , despite our inability to accurately evaluate  $S(0)$  with the present system size,  $S(0) \approx 0.005$ , a value smaller than the computationally determined lower bound recently suggested by de Graff and Thorpe<sup>49</sup> based on studies of continuous random network models. The observed value is comparable to the experimentally determined value for annealed a-Si.<sup>48</sup> Data also show a progressive sharpening of the second neighbor peak in  $g(r)$  associated with the growth of the network peak in  $S(q)$ . Not surprisingly, the region between first and second neighbor shells becomes more and more depleted of particles. The first peak in  $g(r)$  drops off very sharply to a minimum just outside the bonding cutoff distance of  $1 + \delta = 1.12$ .

In Figs. 1(c) and 1(d), we compare  $S(q)$  and  $g(r)$  for the various continuous tetrahedral liquid models to each other. To facilitate comparison, we rescale  $q$  and  $r$  with the position  $r_{\text{peak}}$  of the first peak in  $g(r)$ . The values of  $r_{\text{peak}}$  used in the rescaling are given in Table I. An inspection of the shape of the structure factor near the network peak, i.e.,  $S(q_1)$ , suggests a reasonable similarity between ST2 and TIP4P/2005 with  $\cos \theta = 0.92$ . SW and mW, while similar to each other, have a significantly higher value of  $S(q_1)$  compared to other

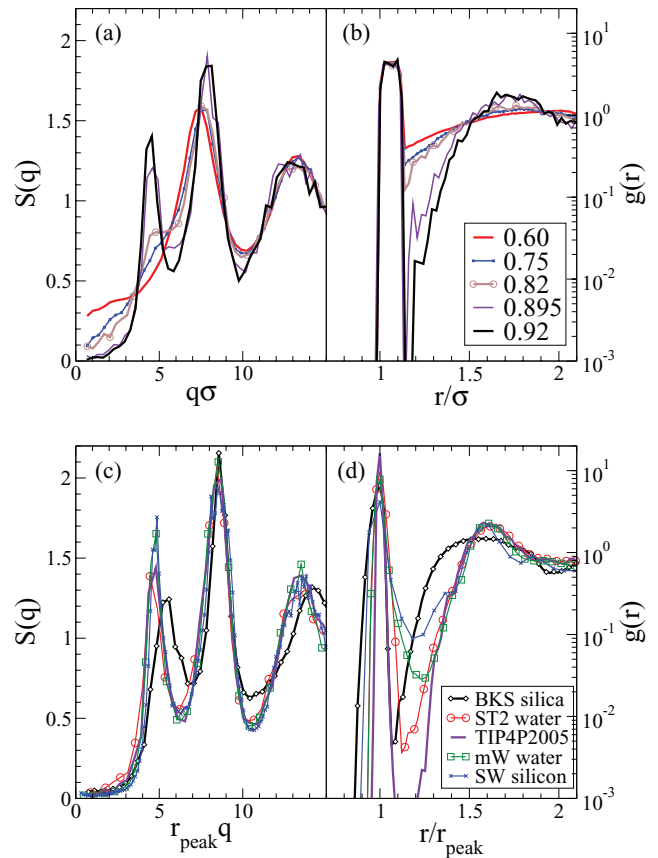


FIG. 1. Structure factor and radial distribution functions. Upper panels give (a)  $S(q)$  and (b)  $g(r)$  for the KF model at  $\rho\sigma^3 = 0.57$ . Legend in (b) gives values of  $\cos \theta$ . There is a progressively higher network peak in  $S(q)$  (near  $q_1 \approx 5$ ) as patches become narrower and a supporting trend in  $g(r)$ , with progressively better defined neighbor coordination shells. Lower panels give (c)  $S(q)$  and (d)  $g(r)$  for the continuous models, for which we rescale distance with  $r_{\text{peak}}$ , the position of the first peak in  $g(r)$ .

models. BKS, however, not only has a significantly lower peak at  $q_1$ , but also has a value of  $q_1$  larger than the common value of  $q_1$  shared by all the other models. Thus, despite having the largest value of  $S(q_2)$ , the network in BKS is distinct from the other models. In terms of  $g(r)$ , the first neighbor peaks of  $\cos \theta = 0.92$ , KF, ST2, and TIP4P/2005 are similar in their sharp fall-off at a similar reduced distance. However, there is a significant difference in the position and width of the second peak, particularly for that of BKS, which is both closer and wider. Thus, it is not surprising that for BKS  $q_1$  is larger and  $S(q_1)$  smaller than for the other models.

We also note that the depth of the minimum at  $r_{\text{cut}}$  in  $g(r)$ , while generally a good indicator of ordering in the liquid, does not uniquely define the number of defects present in the tetrahedral liquid. For example, once we define the neighbors of a particle as those lying with a distance  $r_{\text{cut}}$  of that particle, then a direct calculation of the average number of neighbors reveals that while  $g(r_{\text{cut}})$  for TIP4P/2005 is more than an order of magnitude lower than for BKS, the deviation of the number of neighbors from four is about the same for both models. The average number of neighbors and the fraction of defects for the various models are given in Table I. The fraction of defects is simply the fraction of node particles that do not

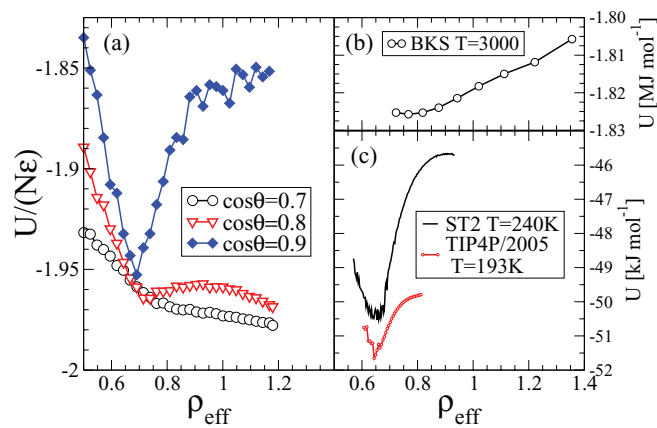


FIG. 2. Potential energy isotherms for (a) the single-bond-per-patch KF model at  $k_B T/\epsilon = 1/9$  (b) BKS silica at  $T = 3000$  K taken from Ref. 37 and (c) ST2 at  $T = 240$  K taken from Ref. 44 and TIP4P/2005 at  $T = 193$ , for which the data are shifted up by  $4.5$  kJ/mol to facilitate comparison with the ST2 model.

have four neighbors. Generally, mW and SW show a relatively large number of defects, and are also both known to crystallize spontaneously in simulation without great difficulty.<sup>19,45</sup>

## B. Energy as a function of density

The scaling of distance with  $r_{\text{peak}}$  allows us to compare optimal network densities by defining a reduced number density  $\rho_{\text{eff}} = \rho r_{\text{peak}}^3$ . Table I shows the value of the reduced density  $\rho_{\text{eff}}^*$  used for structural analysis of the network for the models we study.

Fig. 2(a) shows the potential energy  $U$  of the KF model with the single-bond-per-patch condition enforced at  $k_B T/\epsilon = 1/9$  for  $\cos \theta = 0.70, 0.80$ , and  $0.90$  as a function of  $\rho_{\text{eff}}$ . At  $\cos \theta = 0.90$ , we see a well defined minimum at  $\rho_{\text{eff}} = 0.69$  ( $\rho \sigma^3 = 0.58$ ). At  $\cos \theta = 0.80$ , there is a shallow minimum at a slightly higher value of  $\rho_{\text{eff}}$ , while for  $\cos \theta = 0.70$  there is only a kink in the curve near  $\rho_{\text{eff}} = 0.8$ . The appearance of a minimum in  $U(\rho)$  is thus concurrent with the appearance of the network peak in  $S(q)$ . And as the patch width becomes narrower, this minimum shifts to lower density. We note that for  $\cos \theta = 0.92$  and  $\delta = 0.12$ , the density range for single phase stability for the diamond structure is narrow and occurs near  $\rho \sigma^3 = 0.57$  or  $\rho_{\text{eff}} = 0.68$ . For reference, the number density of the diamond cubic structure for touching hard spheres of unit diameter is  $3\sqrt{3}/8 \approx 0.65$ .

$U(\rho)$  at low  $T$  for the molecular water models in Fig. 2(c) show minima at  $\rho_{\text{eff}} \approx 0.65$ , and while the rise in energy with increasing density is larger for ST2, both models exhibit a significant change in curvature, reaching or approaching a maximum in  $U(\rho)$  near  $\rho_{\text{eff}} \approx 0.80$ – $0.90$ . The BKS model, on the other hand, has a minimum at  $\rho_{\text{eff}} \approx 0.74$ , a significantly larger value than for the other models. While  $U(\rho)$  for BKS does not approach a maximum, the range of data covers the appearance of two inflection points. Once again, we see that the behavior of BKS is significantly different from that of the other models.

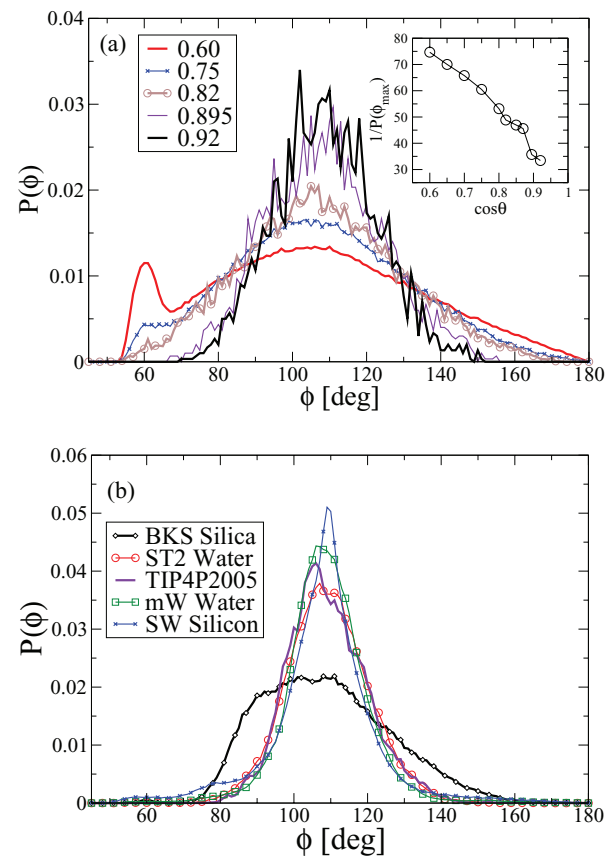


FIG. 3. Probability distribution for the node-node bond angle. Panel (a), KF model: As patches become narrower ( $\cos \theta$  increases), there is a progressive narrowing of  $P(\phi)$ . Values of  $\cos \theta$  are given in the legend. Inset shows a measure of the width of the distribution, the inverse of the maximum probability  $P(\phi_{\text{max}})$ , as a function of  $\cos \theta$ . Panel (b) shows  $P(\phi)$  for the continuous potentials.

## C. Bond angle distributions

In Fig. 3(a) we plot the probability density  $P(\phi)$  for the node-node bond angle  $\phi$  for the KF model for a range of  $\cos \theta$  from  $0.60$  to  $0.92$ . For wide patches, we see a significant peak at  $\phi = 60^\circ$  (arising from a significant number of triangular rings) that disappears by  $\cos \theta \approx 0.82$ , coinciding with the appearance of the network peak in  $S(q)$  and the appearance of a minimum in  $U(\rho)$ . The main peak at the ideal tetrahedral angle of  $109.5^\circ$  grows monotonically with increasing  $\cos \theta$  (narrowing patches), as the whole distribution narrows. We define  $P(\phi_{\text{max}})$  as the maximum value of  $P(\phi)$ , and take as a measure of the width of the distribution the inverse of this height,  $1/P(\phi_{\text{max}})$ , partially to avoid difficulties associated with the peak at  $60^\circ$ . This proxy for the width is shown in the inset to Fig. 3(a) as a function of  $\cos \theta$ . What is clear is that as the patch width narrows, so does the bond angle distribution.

We plot the distribution of angles for the various continuous models in Fig. 3(b) and see that BKS has the broadest distribution. BKS in fact looks to be the odd man out compared to the rest of the models, with a significant portion of its  $P(\phi)$  deviating towards angles smaller than the ideal tetrahedral angle, about which the distributions for the other models are peaked. The tendency of silica to have a broader distribution than for water as well as a smaller average angle was

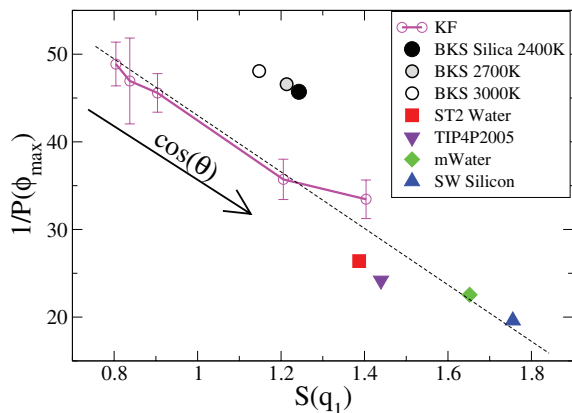


FIG. 4. Inverse of the maximum probability  $P(\phi_{\max})$  of the node-node-node angle distribution as a function of  $S(q_1)$ , the height of the structure factor at the network peak. The values of  $\cos\theta$  for the KF models from left to right are 0.82, 0.85, 0.87, 0.895, and 0.92. Dashed line is a guide to the eye.

already noted in a previous study comparing primitive models of these network formers.<sup>50</sup>

The peaks for the other models become progressively higher in the order of ST2, TIP4P/2005, mW and SW. For mW and SW, the kurtosis of the distribution is quite obviously positive. This non-gaussian shape results in the standard deviation of the distributions not becoming monotonically smaller as the peak height increases, and is another reason why we choose the inverse of peak height as a proxy for the width. Notwithstanding this detail, the heights of the distributions generally shadow the behavior of the height of the network peak in  $S(q)$ .

To make this point more clearly and motivated by the work Yuan and Cormack<sup>51</sup> connecting bond angle distributions and particle correlations beyond first neighbors, we plot in Fig. 4 the quantity  $1/P(\phi_{\max})$  as a function of  $S(q_1)$  for the KF models with  $\cos\theta$  ranging over all the values for which we have a network peak, namely 0.82, 0.85, 0.87, 0.895, and 0.92, along with points corresponding to the continuous models. With the exception of BKS silica, all the models fall near the same line. This linear relationship suggests that all the models (with the exception of silica) belong, in some sense, to the same family of tetrahedral models, and that the members of this family with continuous potentials can be associated, even semiquantitatively, with a KF model with an appropriate patch width. Clearly, the mW and SW models map onto KF models with much narrower patches than those to which ST2 and TIP4P/2005 map. As for BKS, we can only qualitatively say that it would map onto a KF model with even wider patches.

Fig. 4 implies a qualitative similarity in particular between the mW and SW models and the KF models, in that all the data appear to fall quite neatly on the same line. If one thinks of the mW or SW models as patchy particles where the patch geometry is not fixed to a particle axis, then this is not too surprising. What is surprising is just how effectively narrow these patches are.

As a check on BKS silica, we plot in Fig. 4 data points for  $T = 3000$  K (open circle) and  $T = 2700$  K (grey circle) to get a sense of how the values of  $S(q_1)$  and  $1/P(\phi_{\max})$  vary over

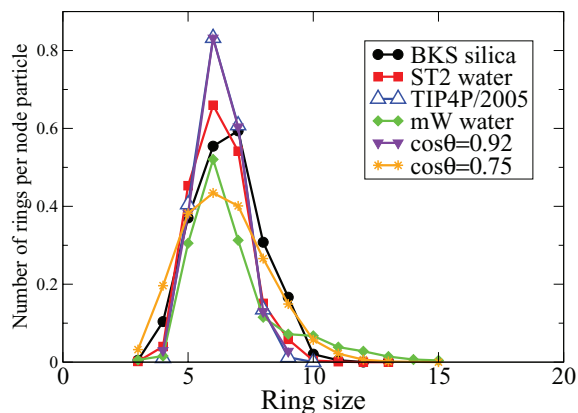


FIG. 5. Ring size distribution for various models.

a small range in  $T$ . We do not see a large variation in either quantity. Neither do we see a trend that might suggest that for some lower  $T$  the data would approach the data from the other models.

#### D. Ring structure

To comment further on the uniqueness of BKS with respect to the other models, we plot in Fig. 5 the size distribution of minimal closed rings formed by neighboring node particles. The KF model, with its tetrahedrally arranged patches, the ST2 and TIP4P/2005 models, wherein tetrahedrality is encoded in the internal angle of the rigid water molecule and the mW model, in which a three-body term enforces a local tetrahedral geometry, all are peaked at a ring size of six, the number expected in crystals at this density. The ring distribution for TIP4P/2005, the most defect-free network we obtain with the water models, is very well reproduced by the  $\cos\theta = 0.92$  KF model. BKS, on the other hand, has a peak at seven, and contains a significant number of rings of size eight and nine. If BKS can be thought of as mapping onto a KF model with wide patches, then it perhaps makes sense that the greater flexibility in node-node “bonds” allows for a broader distribution of ring sizes. However, none of the wider patch KF models yield a similar ring structure, at least not at the density studied. For example, shown in Fig. 5 is the ring size distribution for the  $\cos\theta = 0.75$  KF model, which has a similar number of rings of size eight and nine compared to BKS. However, as is plainly evident, the rest of the curve is quite different.

#### E. KF as ST2 and BKS

The fact that the KF model in the single-particle-per-patch regime, whether enforced or not by the special MD procedure, forms networks with all possible bonds satisfied, allows for the intriguing possibility of using KF configurations as starting points for searching for the lowest energy amorphous configurations in other models. To this end, we study the IS energy of KF configurations after they are converted into ST2 water at  $\rho = 0.83$  g/cm<sup>3</sup> and BKS silica at  $\rho = 2.36$  g/cm<sup>3</sup>.

For ST2, since the rigid molecule already has the same HOH angle as the angle that separates patches on the KF particles, the main complication in converting a fully bonded KF configuration to ST2 is appropriately “filling” only two of four patches of each particle with protons. To do so, we randomly choose a particle that has fewer than two protons assigned and randomly fill with a proton an unfilled patch that is not overlapping with a filled patch from a different particle. This chosen patch is necessarily bonded to another particle in need of at least one proton, and so one of the unfilled patches of this second particle, excluding any patches overlapping with a filled patch (such as the one connecting it back to the first particle), is selected at random to fill. This is repeated until the procedure loops back to the initial particle. The net dipole moment of the loop is nearly zero. The procedure is iterated by randomly selecting a particle to start a new loop until all particles have been visited twice. This method of constructing proton disorder avoids artifacts due to a significant net dipole moment in the simulation cell.<sup>53,54</sup>

After the conversion is made, a CG quench is performed to obtain  $e_{IS}$ . The results are plotted in Fig. 6(a), and show that while at low values of  $\cos \theta$ ,  $e_{IS}$  is rather high, there is a steep drop, with an extrapolated intersection with the average  $e_{IS}$  obtained from simulations of ST2 at  $T = 235$  K, the lowest  $e_{IS}$  that we have, at about  $\cos \theta = 0.94$ . For comparison, in Fig. 6(b) we show  $e_{IS}(T)$  for ST2, with points above  $T = 250$  K taken from Ref. 52. The KF configurations can generally produce ST2 configurations that are quite low in the potential energy landscape.

In order to convert KF into BKS, we take the KF particle positions as the positions of the Si ions and then place O ions at the midway points between bonded KF particles, scaling all coordinates as required. We then perform a CG quench on the system. We have checked other procedures for placing the O ions, for example, placing them midway between the centres of overlapping patches, and have tried quenching only O ion positions first, and then quenching the entire system, but the gain in energy is somewhat marginal. The results of the conversion are plotted in Fig. 6(c). While we do not have sufficient numbers of perfect KF networks at large values of  $\cos \theta$  to make very precise statements, it does not seem improbable to us that the data from lower values of  $\cos \theta$  suggest an extrapolated intersection with our best  $e_{IS}$  for BKS (from  $T = 2400$  K) at a value of  $\cos \theta$  lower than that apparent for ST2, but then rather than achieving the best BKS  $e_{IS}$ , the  $e_{IS}(\cos \theta)$  curve inflects away from the  $T = 2400$  K  $e_{IS}$  line. Such an interpretation would be consistent with the results in Secs. III A–III D that suggest BKS corresponds to a KF model with wider patches than ST2.

For comparison, we plot  $e_{IS}(T)$  for BKS in Fig. 6(d). Two curves are present, one taken from Ref. 37, the other, at lower  $T$ , obtained for the present study. For the new data, we remind the reader that we used the original version of BKS augmented by a potential at very short ranges to prevent “fusion” events and then quenched using the potential used in Ref. 37. That the curves coincide in the region of overlap shows that the additional modifications to BKS in Ref. 37, namely, a fixed Ewald parameter and a tapering of the real space potential energy from 0.77476 nm to zero at 1 nm, do not pro-

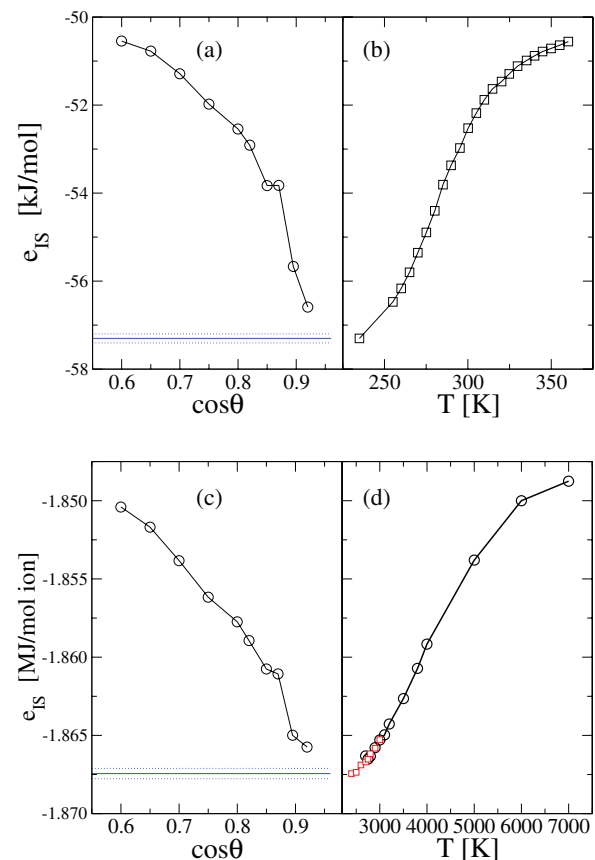


FIG. 6. Energy of inherent structures,  $e_{IS}$ , for (a) ST2 at  $0.83 \text{ g/cm}^3$  and (c) BKS at  $2.36 \text{ g/cm}^3$  generated from fully bonded KF configurations at  $\rho\sigma^3 = 0.57$  with varying patch width. Horizontal lines indicate the  $e_{IS}$  obtained from direct simulations of the original models at the lowest  $T$  probed and same density, with dotted lines indicating standard deviation. Also shown are the full  $e_{IS}(T)$  curves obtained from simulations of (b) ST2 (the point at  $T = 235$  K is obtained in this work, the points at higher  $T$  points are taken from Ref. 52) and (d) BKS (squares indicate data obtained in this work, circles from Ref. 37).

duce significantly different inherent structures from the original BKS potential.

## IV. DISCUSSION

The KF model, augmented with a constraint that ensures only one bond per patch, allows one to unambiguously define a fully bonded energetic ground state even for very wide patches. We study the properties of the ground state of the tetrahedral version of the model as a function of patch width and find that as the patches become narrower, the system becomes progressively more ordered. We mean this in the specific sense that by  $\cos \theta \approx 0.8$ , the system develops a peak in  $S(q)$  characteristic of network-forming liquids, a minimum in the potential energy as a function of density corresponding to an optimal network-forming density and a disappearance of the secondary peak in the bond angle distribution at  $60^\circ$ . The bond angle distribution becomes progressively more peaked as patches become narrower and there is a linear relationship between the width of the angle distribution and the height of the network peak in  $S(q)$ .



By working at optimal network-forming densities with other, continuous models of network-forming liquids such as silica, water and silicon, we remove density as a parameter in an endeavor to roughly map the behavior of these models to the KF model with patch width as the single mapping parameter. As Fig. 4 shows, with the exception of BKS silica, all models studied fall very near the same behavior, and it is thus possible, in principle, to identify the various continuous models with effective KF models of varying patch widths. This is particularly useful in light of recent studies of the KF model pertaining to crystallization<sup>13,14</sup> and liquid stability.<sup>15</sup>

The most tetrahedrally constrained of the models, namely, mW water and SW silicon, correspond to narrow patches. The three-body potential in these models confers a lack of bond flexibility, as do narrow patches in the KF model. The mW and SW models share an important characteristic of the KF models with large  $\cos\theta$ , namely, the difficulty of reaching the energetic ground state before crystallizing. The networks of mW and SW, at the  $T$  we use, are quite imperfect compared to those of BKS, ST2 and TIP4P/2005, and accessing progressively better networks by lowering  $T$  is difficult because of crystallization.

Tetrahedrality in ST2 and TIP4P/2005 is less stringently enforced, and arises from hydrogen bonding between tetrahedrally bent rigid water molecules. Thus, there is more inherent flexibility in the node-node bonds. This observation is consistent with ST2 and TIP4P/2005 mapping to values of  $\cos\theta$  in the range from 0.92 to 0.94, and the fact that nearly defect-free networks within these models are attainable without crystallization.

The network of BKS, in light of the present results, is significantly non-tetrahedral. This is perhaps surprising given the high degree of local structural tetrahedrality.<sup>55</sup> Tetrahedrality in BKS is even less strictly enforced, provided essentially by stoichiometry, steric repulsion and charge. The node-node “bond” is mediated by a relatively large O ion, allowing for a great deal a flexibility. The structural quantities of BKS we study, namely,  $S(q)$ ,  $P(\phi)$ , and ring size distributions are qualitatively different from the other models. Nonetheless, perhaps in a more qualitative way, insights into the the properties of the BKS network can be gained from understanding the properties of KF models with wide patches.

In Ref. 13, the authors showed that the driving force for nucleation, i.e., the chemical potential difference between crystal and liquid  $\Delta\mu$  becomes progressively smaller as  $\cos\theta$  decreases, and a subsequent study of the nucleation barriers enforced the idea that for  $\cos\theta \approx 0.92$  and below, nucleation becomes difficult and the system essentially becomes a glass former. This idea was carried to the extreme in Ref. 15, where the authors showed that if the one-bond-per-patch condition is maintained, then the liquid remains as the stable phase down to  $T = 0$  at the expense of the BCC crystal for a range of densities, certainly for values of  $\cos\theta$  less than or equal to 0.80. This value of 0.80 is interesting in that it also represents the cusp of the KF liquid becoming a networked liquid, rather than simply being four-coordinated.

This idea of increasing patch width to increase liquid stability with respect to crystallization is consistent with previous work done on the SW family of potentials wherein the

three-body constraint strength  $\lambda$  was tuned.<sup>29</sup> At weaker  $\lambda$ , the liquid was stable to progressively lower  $T$ . Were it not for the appearance of the BCC crystal, perhaps the  $T = 0$  limit could be reached.

Obtaining perfect networks for narrow patches is inherently difficult. The rigid geometry enforced by narrow patches or inflexible bonds implies that low energy configurations must resemble crystals, or at least have a reduced number of ways in which a random network can be formed, i.e., a reduction in the configurational entropy and therefore a reduction in liquid stability. If one wished to map the KF model to models such as mW and SW in a more precise way in spite of this difficulty, one could perhaps compare the state points with similar defect concentration.

## V. CONCLUSIONS

We study a family of fully bonded tetrahedral KF patchy particle systems as a function of patch width at the optimal network-forming density and find a few concurrent measures for the onset of a structured network at  $\cos\theta \approx 0.80$ . There is a linear relationship between the width of the bond angle distribution and the height of the network peak in the structure factor. Several other models follow this “family line,” including the ST2, TIP4P/2005, and mW models of water, as well as SW silicon. This suggests a mapping of these models to KF models of different patch widths. This mapping makes intuitive sense given the nature of the potentials and degree of bond flexibility in each of the models.

The mapping is useful given the systematic study of how the patch width affects the ability of the KF model to crystallize, or conversely, to avoid crystallization. Essentially, the narrower the patches, the greater the propensity for the model to crystallize. Wider patches allow for the system to approach or even reach the ground state energy and avoid nucleation.<sup>12–15</sup> According to the semi-qualitative mapping, SW and mW map to quite narrow patches, and this explains why they are prone to crystallize before achieving a relatively defect-free network. ST2 and TIP4P/2005, on the other hand, map to wider patches and are therefore better network glass-formers. The perfect correspondence between crystallization ability and bonding angular width observed throughout all these models reinforces the general validity of the results obtained in the investigation of the KF model.

BKS silica, the best glass former, is qualitatively different in terms of several properties studied and does not fall near the family line. Its network properties are rather different. Possibly, one needs to devise a binary mixture analog<sup>50,56–58</sup> in order to capture the essential differences. A form of the KF model has also been recently shown to have a liquid that is the thermodynamic ground state for a range of densities for sufficiently wide patches.<sup>15</sup> It remains a challenge to alter the continuous molecular potentials in order to achieve an analogous liquid ground state.

Finally, we like to note that man-made particles can help shed light on unsolved problems in atomic and molecular physics, in the present case connecting gel-forming patchy colloids and tetrahedral network glass formers.<sup>59</sup> New soft-matter systems in which valence can be precisely controlled,

e.g., DNA constructs<sup>5,60</sup> and new polymers,<sup>61</sup> may contribute to deepening our understanding of fundamental problems in disordered systems. Providing valence to colloids<sup>4,62</sup> is opening a very rich line of investigation.

## ACKNOWLEDGMENTS

I.S.-V. thanks NSERC and ERC-PATCHYCOLLOIDS for funding, ACEnet for computational support, CFI for funding of computing infrastructure, and Sapienza University for hosting. Both F.S. acknowledge support from ERC-226207-PATCHYCOLLOIDS and MIUR-PRIN.

- <sup>1</sup>K. Binder and W. Kob, *Glassy Materials And Disordered Solids: An Introduction to Their Statistical Mechanics* (World Scientific Publishing Company, 2005).
- <sup>2</sup>A. B. Pawar and I. Kretschmar, *Macromol. Rapid Commun.* **31**, 150 (2010).
- <sup>3</sup>E. Bianchi, R. Blaak, and C. N. Likos, *Phys. Chem. Chem. Phys.* **13**, 6397 (2011).
- <sup>4</sup>Y. Wang, Y. Wang, D. R. Breed, V. N. Manoharan, L. Feng, A. D. Hollingsworth, M. Weck, and D. J. Pine, *Nature (London)* **491**, 51 (2012).
- <sup>5</sup>S. Biffi, R. Cerbino, F. Bomboi, E. M. Paraboschi, R. Asselta, F. Sciortino, and T. Bellini, *Proc. Natl. Acad. Sci. U.S.A.* **110**, 15633 (2013).
- <sup>6</sup>Q. Chen, S. C. Bae, and S. Granick, *Nature (London)* **469**, 381 (2011).
- <sup>7</sup>L. Vlcek and I. Nezbeda, *Mol. Phys.* **102**, 771 (2004).
- <sup>8</sup>I. Nezbeda, *Mol. Phys.* **103**, 59 (2005).
- <sup>9</sup>J. Kolafa and I. Nezbeda, *Mol. Phys.* **61**, 161 (1987).
- <sup>10</sup>N. Kern and D. Frenkel, *J. Chem. Phys.* **118**, 9882 (2003).
- <sup>11</sup>W. Bol, *Mol. Phys.* **45**, 605 (1982).
- <sup>12</sup>F. Romano, E. Sanz, and F. Sciortino, *J. Chem. Phys.* **132**, 184501 (2010).
- <sup>13</sup>F. Romano, E. Sanz, and F. Sciortino, *J. Chem. Phys.* **134**, 174502 (2011).
- <sup>14</sup>I. Saika-Voivod, F. Romano, and F. Sciortino, *J. Chem. Phys.* **135**, 124506 (2011).
- <sup>15</sup>F. Smallenburg and F. Sciortino, *Nat. Phys.* **9**, 554 (2013).
- <sup>16</sup>B. W. H. van Beest, G. J. Kramer, and R. A. van Santen, *Phys. Rev. Lett.* **64**, 1955 (1990).
- <sup>17</sup>F. H. Stillinger and A. Rahman, *J. Chem. Phys.* **60**, 1545 (1974).
- <sup>18</sup>J. L. F. Abascal and C. Vega, *J. Chem. Phys.* **123**, 234505 (2005).
- <sup>19</sup>V. Molinero and E. B. Moore, *J. Phys. Chem. B* **113**, 4008 (2009).
- <sup>20</sup>F. H. Stillinger and T. A. Weber, *Phys. Rev. B* **31**, 5262 (1985).
- <sup>21</sup>C. De Michele, S. Gabrielli, P. Tartaglia, and F. Sciortino, *J. Phys. Chem. B* **110**, 8064 (2006).
- <sup>22</sup>W. H. Zachariasen, *J. Am. Chem. Soc.* **54**, 3841 (1932).
- <sup>23</sup>D. L. Evans and S. V. King, *Nature (London)* **212**, 1353 (1966).
- <sup>24</sup>R. J. Bell and P. Dean, *Nature* **212**, 1354 (1966).
- <sup>25</sup>M. G. Sceats, M. Stavola, and S. A. Rice, *J. Chem. Phys.* **70**, 3927 (1979).
- <sup>26</sup>S. A. Rice and M. G. Sceats, *J. Chem. Phys.* **85**, 1108 (1981).
- <sup>27</sup>A. R. Henn and W. Kauzmann, *J. Phys. Chem.* **93**, 3770 (1989).
- <sup>28</sup>D. E. Polk and D. S. Boudreaux, *Phys. Rev. Lett.* **31**, 92 (1973).
- <sup>29</sup>V. Molinero, S. Sastry, and C. A. Angell, *Phys. Rev. Lett.* **97**, 075701 (2006).
- <sup>30</sup>F. Wooten, K. Winer, and D. Weaire, *Phys. Rev. Lett.* **54**, 1392 (1985).
- <sup>31</sup>G. T. Barkema and N. Mousseau, *Phys. Rev. B* **62**, 4985 (2000).
- <sup>32</sup>N. Mousseau and G. T. Barkema, *J. Phys.: Condens. Matter* **16**, S5183 (2004).
- <sup>33</sup>F. H. Stillinger, *J. Chem. Phys.* **88**, 7818 (1988).
- <sup>34</sup>F. H. Stillinger, *Science* **267**, 1935 (1995).
- <sup>35</sup>F. Sciortino, *J. Stat. Mech.* **2005**, P05015.
- <sup>36</sup>J. Horbach and W. Kob, *Phys. Rev. B* **60**, 3169 (1999).
- <sup>37</sup>I. Saika-Voivod, F. Sciortino, and P. H. Poole, *Phys. Rev. E* **69**, 041503 (2004).
- <sup>38</sup>I. Saika-Voivod, F. Sciortino, T. Grande, and P. H. Poole, *Philos. Trans. R. Soc. London, Ser. A* **363**, 525 (2005).
- <sup>39</sup>H. J. C. Berendsen, D. van der Spoel, and R. van Druren, *Comput. Phys. Commun.* **91**, 43 (1995).
- <sup>40</sup>E. Lindahl, B. Hess, and D. van der Spoel, *J. Mol. Model.* **7**, 306 (2001).
- <sup>41</sup>D. van der Spoel, E. Lindahl, B. Hess, G. Groenhof, A. E. Mark, and H. J. C. Berendsen, *J. Comput. Chem.* **26**, 1701 (2005).
- <sup>42</sup>B. Hess, C. Kutzner, D. van der Spoel, and E. Lindahl, *J. Chem. Theory Comput.* **4**, 435 (2008).
- <sup>43</sup>J. L. F. Abascal and C. Vega, *J. Chem. Phys.* **133**, 234502 (2010).
- <sup>44</sup>F. Sciortino, I. Saika-Voivod, and P. H. Poole, *Phys. Chem. Chem. Phys.* **13**, 19759 (2011).
- <sup>45</sup>V. V. Vasisht, S. Saw, and S. Sastry, *Nat. Phys.* **7**, 549 (2011).
- <sup>46</sup>X. Yuan and A. N. Cormack, *Comput. Mater. Sci.* **24**, 343 (2002).
- <sup>47</sup>S. R. Elliott, *Nature (London)* **354**, 445 (1991).
- <sup>48</sup>R. Xie, G. G. Long, S. J. Weigand, S. C. Moss, T. Carvalho, S. Roorda, M. Hejna, S. Torquato, and P. J. Steinhardt, *Proc. Natl. Acad. Sci. U.S.A.* **110**, 13250 (2013).
- <sup>49</sup>A. M. R. de Graff and M. F. Thorpe, *Acta Crystallogr. A* **66**, 22 (2010).
- <sup>50</sup>C. De Michele, P. Tartaglia, and F. Sciortino, *J. Chem. Phys.* **125**, 204710 (2006).
- <sup>51</sup>X. Yuan and A. N. Cormack, *J. Non-Cryst. Solids* **319**, 31 (2003).
- <sup>52</sup>P. H. Poole, S. R. Becker, F. Sciortino, and F. W. Starr, *J. Phys. Chem. B* **115**, 14176 (2011).
- <sup>53</sup>A. Rahman and F. H. Stillinger, *J. Chem. Phys.* **57**, 4009 (1972).
- <sup>54</sup>V. Buch, P. Sandler, and J. Sadlej, *J. Phys. Chem. B* **102**, 8641 (1998).
- <sup>55</sup>M. S. Shell, P. G. Debenedetti, and A. Z. Panagiotopoulos, *Phys. Rev. E* **66**, 011202 (2002).
- <sup>56</sup>M. H. Ford, S. M. Auerbach, and P. A. Monson, *J. Chem. Phys.* **121**, 8415 (2004).
- <sup>57</sup>E. Bianchi, P. Tartaglia, and F. Sciortino, *J. Chem. Phys.* **129**, 224904 (2008).
- <sup>58</sup>I. Saika-Voivod, H. M. King, P. Tartaglia, F. Sciortino, and E. Zaccarelli, *J. Phys.: Condens. Matter* **23**, 285101 (2011).
- <sup>59</sup>F. Sciortino, *Eur. Phys. J. B* **64**, 505 (2008).
- <sup>60</sup>Y. Li, Y. D. Tseng, S. Y. Kwon, L. d'Espaux, J. S. Bunch, P. L. McEuen, and D. Luo, *Nature Mater.* **3**, 38 (2004).
- <sup>61</sup>D. Montarnal, M. Capelot, F. Tournilhac, and L. Leibler, *Science* **334**, 965 (2011).
- <sup>62</sup>G. Zhang, D. Wang, and H. Möhwald, *Angew. Chem., Int. Ed.* **44**, 7767 (2005).